Chapter 1

Nano-scale testing of nanowires and carbon nanotubes using a microelectromechanical system

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The need to characterize nanometer-scale materials and structures has grown tremendously in the past decade. These structures may behave very differently from their larger counterparts and must be carefully characterized before their full potential is realized. The challenging task of mechanical characterization requires an entirely new set of techniques to achieve the force and displacement resolution needed to accurately characterize these structures. This chapter begins with a brief review of some of the methods used in mechanical characterization of nano-scale specimens, followed by a detailed description of a MEMS-based material testing system. This MEMS-based system allows for continuous observation of specimen deformation and failure with sub-nanometer resolution by scanning or transmission electron microscope while simultaneously measuring the applied load electronically with nano-Newton resolution. Special emphasis is placed on modeling and analysis of a thermal actuator used to apply a displacement-controlled load to the tensile specimen as well as the electrostatic load sensor. Finally, experimental results demonstrating the advantages of the MEMS-based system are presented.

1.1. Introduction

The emergence of numerous nano-scale materials and structures within the past decade has prompted a need for methods to characterize their

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unique mechanical properties. For example, nanotubes and nanowires are seen as ideal structures for use in a variety of applications ranging from nanocomposites to nanoelectromechanical systems. However, their scale presents a new set of challenges to the mechanics community. Identification of their properties and deformation mechanisms requires techniques of loading, measuring, and imaging with finer resolutions than previously achieved. As a result, these structures demand quantitative *in situ* mechanical testing by scanning or transmission electron microscope, or scanning probe microscope.

1.2. Mechanical characterization techniques

The precision required in nano-scale material testing is prohibitive to many techniques used previously on larger scales. This is due to strict requirements in: handling, manipulating, and positioning specimens; applying and measuring forces in the nano-Newton range, and; measuring local deformation. Existing material testing techniques intended to overcome these limitations can be roughly categorized into three types: (1) dynamic vibration, (2) bending, and (3) tensile tests.

1.2.1. Dynamic vibration

The Young’s modulus of nanostructures can be estimated by observing their vibrations. Treacy et al.\(^1\) determined the Young’s modulus of multi-walled carbon nanotubes (MWNTs) by measuring the amplitude of their thermal vibrations within a transmission electron microscope (TEM). One end of the MWNT was attached to the edge of a nickel ring while the other end remained free. The ring and MWNT were then imaged in the TEM. The frequency of the thermal vibration of the MWNT was significantly faster than the integration time required for imaging, causing the free end to appear blurred. The authors used the blurred region to determine an envelope of vibration. This envelope increased significantly with temperature, indicating it was indeed a thermal vibration. The Young’s modulus of the MWNT was estimated based on the size of the envelope.

Poncharal et al.\(^2\) measured the Young’s modulus of MWNTs by inducing resonance within a TEM. Here the MWNTs were attached to a gold wire and an electrode was introduced using a piezo-driven translation stage on the TEM holder. An AC voltage was then applied across the wire and electrode, causing oscillatory deflection of the MWNTs toward the elec-
trode. By increasing the driving frequency to the point of resonance, the authors were able to estimate the Young’s modulus based on the measured geometry of the MWNTs and their resonance frequency.

1.2.2. Bending

Bending techniques, including force spectroscopy atomic force microscopy (AFM), nanoindentation, and on-chip testing, involve application of a known bending force while measuring the resulting displacement. These techniques typically lack the ability to image the specimen during loading. Wong et al. measured the Young’s modulus, strength, and toughness of MWNTs and SiC nanorods using an atomic force microscope (AFM). The nanostructures were randomly dispersed on a flat substrate and pinned in place by microfabricated patches. The AFM was then used to bend the cantilevered structures transversely. By measuring the lateral force applied to the AFM probe by the nanostructure, the authors were able to obtain force versus deflection data at various locations along the length of the structure.

In the above method, adhesion and friction between the nanostructure and substrate could not be avoided. To avoid these issues, Walters et al. suspended MWNTs over a microfabricated trench before bending them laterally with an AFM. Salvetat et al. dispersed MWNTs over an alumina ultrafiltration membrane with 200 nm pores. This created similarly suspended nanostructures. The natural adhesion between the MWNTs and membrane was found to be sufficiently strong to fix the MWNTs during testing. The authors then deflected the suspended MWNTs vertically using an AFM probe in contact mode to obtain similar force-displacement measurements.

1.2.3. Tensile tests

The tensile test is perhaps the most direct method of determining the Young’s modulus of a material. Some tensile techniques allow for simultaneous load measurement and local imaging by AFM, optical interferometry, or scanning electron microscopy (SEM). On a larger scale, Pan et al. used a stress-strain rig to load a long (approximately 2 mm) MWNT rope in tension. This rope contained tens of thousands of parallel nanotubes. Sharpe et al. loaded thin films in tension while simultaneously measuring displacement using either a capacitance-based displacement probe or laser interferometry, depending on the size of the sample. To
use laser interferometry, two closely-spaced reflective gauge markers were patterned on the specimen. When shining a laser on these markers, interference fringes form which move as the spacing between the markers changes. These techniques tend to be better suited to the micrometer scale and not to the study of nanostructures such as nanotubes or nanowires due to their limited resolution.

On a smaller scale, Yu et al.\textsuperscript{12} and Ding et al.\textsuperscript{13} used a micro- or nanomanipulator to conduct in situ SEM tensile testing of MWNTs. The authors fixed a single nanotube between two AFM probes by localized electron beam-induced deposition (EBID) of carbonaceous material within the SEM chamber. One of the AFM probes was of low stiffness (less than 0.1 N/m) and used as a load sensor. The other probe was rigid and used to apply tensile load. When the rigid probe was actuated, the soft probe deflected in response to the applied load. The force applied to the nanotube was estimated based on the deflection of the soft cantilever and its known stiffness. The deformation of the nanotube was recorded by the SEM. The Young’s modulus and failure strength of the nanotubes were then calculated based on the applied forces and corresponding measured displacements.

Marszalek et al.\textsuperscript{3} attached a gold nanowire to an AFM probe of known stiffness. By lifting the probe a prescribed amount using the piezoelectric actuators of the AFM and observing the corresponding deflection of the probe, they were able to deduce the force applied to the nanowire and corresponding displacement.

Other techniques combine microfabricated and larger-scale devices. These allow alternating SEM or TEM imaging and load or deformation measurement modes through switching of the imaging electron beam between the specimen and microfabricated beams used as load sensors.\textsuperscript{15,16} During this switching, important local deformation events may go unobserved. Finally, some techniques have been developed to provide real time images of the specimen during loading. However, quantitative measurement of load and deformation are not provided simultaneously.\textsuperscript{17,18}

The methods described above represent some of the significant progress made recently in the mechanical testing of nanostructures. However, lack of control in experimental conditions or limited accuracy of force and displacement measurements can limit their applicability. Recent advances in microelectromechanical systems (MEMS) create the potential for material testing systems that overcome some of the described limitations.
1.3. A MEMS-based material testing stage

MEMS lend themselves naturally to material testing at the nanometer scale. These systems consist of combinations of micromachined elements, including strain sensors and actuators, integrated on a single chip. Due to their intermediate size, MEMS serve as an excellent interface between the macro and nano world. Their extremely fine force and displacement resolution allows accurate measurement and transduction of forces and displacements relevant at the nanometer scale. At the same time, the larger feature sizes and signal levels of MEMS allow handling and addressing by macro-scale tools. Furthermore, many of the sensing and actuation schemes employed in MEMS scale favorably. For example, the time response, sensitivity, and power consumption of electrostatic displacement sensors improves as their dimensions shrink.

Electrostatic comb-drive actuators are often used in MEMS-based testing systems to apply time-dependent forces. van Arsdell and Brown\textsuperscript{19} repeatedly stressed a micrometer-scale specimen in bending using a comb-drive actuator fixed to one end of the specimen. This comb-drive actuator swept in an arc-like motion while the opposite end of the specimen was fixed, causing bending stresses in the specimen. The comb-drive was also used to measure displacement, allowing for fracture and fatigue data to be collected when testing to the point of failure. Kahn, \textit{et al.}\textsuperscript{20,21} determined fracture toughness by controlling crack propagation in a notched specimen using a comb-drive. One end of the specimen was fixed while the other was attached to a perpendicularly oriented comb-drive.

Electrothermal actuation schemes have also been used to apply loading.\textsuperscript{22,23} In these actuators, Joule heating induces localized thermal expansion of regions of the actuators and an overall displacement. The resulting strains are often measured using an integrated capacitive sensor and may be verified through digital image correlation.

This section presents a detailed description of the design and modeling of a MEMS-based material testing system\textsuperscript{24} for in-situ electron microscopy mechanical testing of nanostructures. This device allows for continuous observation of specimen deformation and failure with sub-nanometer resolution by SEM or TEM while simultaneously measuring the applied load electronically with nano-Newton resolution. To begin, an analytical model of the thermal actuator used to apply tensile loading includes an electrothermal analysis to determine the temperature distribution in the actuator, followed by a thermomechanical analysis to determine the resulting displace-
ment. A coupled-field finite element analysis complements the analytical model. Next, the differential capacitive sensor is analyzed to determine the applied load from the measured electric signals. Finally, a set of design criteria are established based on the analyses as guidelines for design of similar devices.

1.3.1. **Device description**

![Fig. 1.1](image.png)

Fig. 1.1. Two variations of the MEMS-based material testing stage. (a) “Displacement controlled” device using a thermal actuator and differential capacitive load sensor. (b) “Force controlled” device using an electrostatic comb-drive actuator and differential capacitive load sensor.

The MEMS-based tensile loading stage \(^{24–26}\) consists of a linear actuator and a load sensor with a specimen fixed between them. The actuator is used to apply a tensile load to a specimen attached between the actuator and load sensor, while the load sensor detects the corresponding load. Fig. 1.1a shows the entire device. The electrothermal actuator acts as a “displacement control” in the sense that it applies a prescribed displacement to the specimen regardless of the force required to achieve this displacement (within the functional range of the device). The load sensor is suspended on a set of folded beams of known stiffness and measures the corresponding tensile force applied to the specimen. Fig. 1.1b shows an alternative loading stage using an electrostatic rather than a thermal actuator. The electrostatic actuators works as a “force control”, applying a prescribed force regardless of the resulting displacement (again within a functional range).

While both the thermal and electrostatic actuators lend themselves nicely to standard microfabrication techniques, the remainder of this chapter focuses on the device using the thermal actuator as a case study in the design and modeling involved in building such a device. Electrostatic
1.3.2. Electrothermal actuator

Electrothermal actuation compliments electrostatic schemes as a compact, stable, high-force actuation technique.\textsuperscript{29} It involves coupling of electric, thermal, and structural fields. Typically a resistive heating element is used as a heat input. This invokes thermal expansion of the device, resulting in a displacement. As mentioned above, these actuators are considered a “displacement control” as they displace a specific amount for a given heat input (within the limits of buckling and material stiffness).

Various forms of thermal actuators have been employed in systems ranging from linear and rotary microengines,\textsuperscript{30} to two-dimensional nano-scale positioners,\textsuperscript{31} optical benches,\textsuperscript{32} and instrumentation for material characterization.\textsuperscript{33} By incorporating compliant mechanisms, larger displacements can be achieved.\textsuperscript{31}

Modeling of thermal actuators generally takes one of two approaches:

(1) A sequential electro-thermal and thermo-structural analysis,\textsuperscript{34–36} or;
(2) A complete coupled three-dimensional finite element analysis (FEA).\textsuperscript{37}

Additional analyses include characterization of the temperature-dependent electro-thermal properties\textsuperscript{29,37} of these devices.

Sections 1.3.3 and 1.3.4 present an analysis of the thermal actuator employed by Zhu and Espinosa\textsuperscript{24,25} in the MEMS-based material testing system. The description begins with derivation of a set of analytical expressions for the response of the thermal actuator using a structural mechanics approach. This analysis is followed by a three-dimensional finite element multiphysics simulation to assess the temperature distribution within the actuator.

1.3.3. Analytical modeling of the thermal actuator

A schematic of the thermal actuator to be analyzed is shown in Fig. 1.2. The thermal actuator consists of a series of inclined polysilicon beams supporting a free-standing shuttle. One end of each of the inclined beams is anchored to the substrate while the opposite end connects to the shuttle. Thermal expansion of the inclined beams, induced by Joule heating, causes the shuttle to move forward. This heating is the result of current flowing through the beams driven by a voltage applied across the two anchor
Fig. 1.2. (a) Schematic of the thermal actuator. (b) Cross section of a single beam suspended over the substrate.

points.\textsuperscript{38} Modeling of these actuators requires a two-step analysis; first an electrothermal analysis to determine the temperature distribution in the device, followed by a thermostructural analysis to determine the resulting displacement field.

1.3.3.1. \textit{Electrothermal model}

An electrothermal model of the device is developed to determine the temperature distribution as a function of the applied voltage. This is highly dependent upon the operating environment. When operating in air, the dominant heat transfer mechanism is heat conduction between the actuator and substrate through the air-filled gap between them.\textsuperscript{35,36,38} In contrast, heat dissipation by conduction through the anchors to the substrate dominates in vacuum.\textsuperscript{35,38} Assuming each beam is thermally independent, an electrothermal model based on a single beam is presented.\textsuperscript{35} Heat transfer within the beam is treated as a one-dimensional problem since the length dimension is significantly larger than either of the cross-sectional dimensions.

First examine the case in air, where heat conduction through the air-filled gap between the actuator and substrate is the dominant mechanism of heat transfer. Here the governing equation is,

\[
k_p \frac{d^2T}{dx^2} + J^2 \rho = \frac{S}{h} \frac{T - T_s}{R_T},
\]

where \(k_p\) and \(\rho\) are the thermal conductivity and resistivity respectively of the polysilicon beams; \(J\) is the current density; \(S = \frac{h}{w} \left( \frac{2h}{h_{air}} + 1 \right) + 1\) is a shape factor accounting for the effect of element shape on heat conduction.
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to the substrate; \( R_T = \frac{h_{air}}{k_{air}} + \frac{h_n}{k_n} + \frac{h_s}{k_s} \) is the thermal resistance between
the polysilicon beam and substrate; \( h \) and \( w \) are the thickness and width
of a single beam respectively; \( h_{air} \) is the gap between the beam and silicon
nitride layer on the substrate; \( h_n \) is the thickness of the silicon nitride; \( h_s \)
is the representative thickness of the substrate; \( k_{air}, k_n, k_s \) are the thermal
conductivities of air, silicon nitride, and the substrate respectively; and \( T_s \)
is the temperature of the substrate.

The thermal conductivities \( k_p \) and \( k_{air} \) are both temperature dependent.
However, the assumption of a constant \( k_p \) yields results similar to those
using a temperature-dependent value of \( k_p \). Assuming a constant \( k_p \) and
temperature dependent \( k_{air} \), the finite difference method is implemented
to solve (1.1) by writing the second-order differential equation in the form
\[
\frac{d^2 T}{dx^2} = b(x, T),
\]
and approximating it as,
\[
\frac{d^2 T}{dx^2} \approx \frac{1}{(\Delta x)^2} (T_{i+1} - 2T_i + T_{i-1}).
\]

Fig. 1.3. Steady state temperature profile (with respect to the substrate) along a pair
of inclined beams and the shuttle operated in air with an input current of 10 mA for
both constant and temperature-dependent values of \( k_{air} \). Locations 0-300 \( \mu m \) and 360-
660 \( \mu m \) correspond to the beams while locations 300-360 \( \mu m \) correspond to the shuttle
between the beams. The beams are anchored to the substrate at locations 0 and 660
\( \mu m \).
Fig. 1.3 shows the steady-state temperature profile obtained for a two-leg (one pair of inclined beams) thermal actuator operating in air. The temperature of the shuttle is significantly lower than that of the majority of each beam. This is due to the relatively low current density in the shuttle, resulting in a lower rate of heat generation as compared to that of the beams. Furthermore, the relatively large area of the shuttle results in greater heat dissipation through the air to the substrate.

The thermal conductivity of the air has a significant effect on the actuator behavior. This strong dependence is clearly seen in Fig. 1.3, where the only difference between the two curves is the temperature dependence of the thermal conductivity of air, \( k_{\text{air}} \). Increasing the heat flow between the beams and shuttle and the substrate. Consequently, the temperature of the beams and shuttle is lower for a given current flow. Clearly, decreasing heat conduction through the air increases the temperature of the beams. Ultimately, operation in vacuum maximizes the beam temperature for a given current flow, making the device more efficient.

To analyze the case where the thermal actuator operates in vacuum, remove the term for heat conduction through the air from (1.1),

\[
k_p \frac{dT}{dx^2} + J^2 \rho = 0.
\]

Fig. 1.4 shows that the highest temperature now occurs in the shuttle rather than in the beams. Here the temperature depends most upon the distance from the anchor points which are now assumed to be the only source of heat dissipation. Since the shuttle is furthest from the anchors, it reaches the highest temperature.

1.3.3.2. Thermomechanical model

With the temperature distribution now known from the electrothermal analysis, the thermomechanical behavior of the actuator is modeled to determine the resulting displacement. The following assumptions are made in the analytical derivation of the thermomechanical behavior:

- the average temperature increase in the inclined beams is known;
- deformation of the central shuttle is negligible compared to that of the inclined beams;
- all strains and displacements are small, and;
- there is negligible shear deformation of the beams.
Fig. 1.4. Steady state temperature profile along a pair of inclined beams and the shuttle operated in vacuum with an input current of 3 mA. The thermal conductivity of polysilicon is assumed temperature dependent. Locations 0-300 µm and 360-660 µm correspond to the beams while locations 300-360 µm correspond to the shuttle between the beams. The beams are anchored to the substrate at locations 0 and 660 µm.

As in the electrothermal analysis, a single pair of inclined beams is first considered as shown in Fig. 1.5a. Later the entire device, including the thermal actuator, specimen, and load sensor is analyzed.

Fig. 1.5. Schematic of a pair of inclined beams subjected to an average increase in temperature $\Delta T$. (a) Two beams joined at the central shuttle; (b) Equivalent mechanical representation of a single beam.
The pair of inclined beams, forming a single V-shaped clamped beam, is subjected to a uniform increase in temperature along its length. While the thermomechanical response of a similar structure was previously approximated, the following analysis follows a rigorous structural mechanics approach.

![Schematic of an inclined beam in a local reference frame.](image)

Exploiting the symmetry of the system, the mechanical response is equivalently computed considering half of the structure as shown in Fig. 1.5b. To determine the axial force in the beam and the displacement of Node A (Fig. 1.5) in the $y$-direction, the elastic stiffness matrix of the beam is assembled relative to the displacements in Node A. Before obtaining the stiffness matrix in the global frame, it is first computed in a local reference frame as shown in Fig. 1.6. The governing structural equations for the beam subjected to an average temperature increase $\Delta T$ are,

$$
\begin{bmatrix}
\frac{EA}{l} & 0 \\
0 & 12EI
\end{bmatrix}
\begin{bmatrix}
U_A^\xi \\
U_A^\eta
\end{bmatrix}
= \begin{bmatrix}
\alpha T E A \\
0
\end{bmatrix}
+ \begin{bmatrix}
R_A^\xi \\
R_A^\eta
\end{bmatrix},
$$

where $E$, $A$, and $l$ are the Young’s modulus, cross-sectional area, and length of the beam respectively; $I$ is the moment of inertia of the beam with respect to the out-of-plane axis $\zeta$ in the local reference frame; $U_A^\xi$ and $U_A^\eta$ are the displacements of Node A in the directions $\xi$ and $\eta$ respectively; $\alpha$ is the coefficient of thermal expansion of the beam material; and $R_A^\xi$ and $R_A^\eta$ are the reaction forces at Node A in directions $\xi$ and $\eta$ respectively.

The boundary conditions are known in terms of the global $x$-$y$ reference frame in Fig. 1.5. Thus (1.2) is transformed to the global system by a rotation matrix relating the local and global degrees of freedom,

$$
\begin{bmatrix}
U_A^\xi \\
U_A^\eta
\end{bmatrix}
= \begin{bmatrix}
\cos \theta & \sin \theta \\
-\sin \theta & \cos \theta
\end{bmatrix}
\begin{bmatrix}
U_A^x \\
U_A^y
\end{bmatrix}.
$$
Applying this relation, (1.2) becomes,
\[
\left[ c^2 \frac{EA}{l} + s^2 \frac{12EI}{l^3} \right] U_x^A + \left[ \frac{EA}{l} - \frac{12EI}{l^3} \right] \left( \frac{EA}{l} + c \right) U_y^A = \left[ \alpha \Delta T E A c \right] + \left[ R_x^A \right],
\]
where \( c = \cos \theta \) and \( s = \sin \theta \). The boundary conditions reflecting the constraint at Node A are \( U_x^A = 0 \), \( R_x^A \neq 0 \), \( U_y^A \neq 0 \), and \( R_y^A = 0 \). Applying these to (1.3) yields the reaction force on Node A in the \( x \)-direction and the displacement of Node A in the \( y \)-direction for an average increase in temperature \( \Delta T \) along the beam, namely;
\[
R_x^{\Delta T} = R_x^A = -\alpha \Delta T E A \frac{c}{s^2 \frac{12EI}{l^3} + c^2} \equiv -\alpha \Delta T E A \frac{c}{s^2 \Psi + c^2},
\]
\[
U_y^{\Delta T} \equiv U_y^A = \alpha \Delta T I \frac{s}{s^2 + c^2 \frac{12EI}{l^2}} \equiv \alpha \Delta T I \frac{s}{s^2 + \frac{\Psi c^2}{\Psi}}. \tag{1.4}
\]
Here the dimensionless parameter \( \Psi = AI^2/12I \) represents the ratio of the axial and bending stresses. Using the reaction force \( R_x^A \), it is possible to obtain the compressive axial force \( N \) in the beam by projection along the axial direction, \( N = R_x^A c \). A similar analysis yields the response of a pair of inclined beams subjected to an external force \( F \) applied to the central shuttle acting in the \( y \)-direction. In this case, the axial internal force and displacement of Node A are determined using the same system of governing equations (1.3) by replacing the vector on the right hand side that depends on the temperature increase with the external applied force vector \( [0 \ F/2]^T \). The axial force and displacement now become,
\[
R_x^F \equiv R_x^A = cs \left( \frac{EA}{l} - \frac{12EI}{l^3} \right) U_y^A = F \frac{cs (\Psi - 1)}{2(s^2 \Psi + c^2)},
\]
\[
U_y^F \equiv U_y^A = F \left( \frac{1}{2(s^2 \frac{12EI}{l^2} + c^2 \frac{12EI}{l^3})} \right) = F I \frac{1}{EA} \frac{1}{2(s^2 + \frac{\Psi c^2}{\Psi})}.
\]
Using the displacement \( U_y^F \) for a given force \( F \), the stiffness of the V-shaped clamped thermal beam shown in Fig. 1.5 is,
\[
K_{tb} \equiv \frac{F}{U_y^F} = 2 \left( s^2 + \frac{c^2}{\Psi} \right) \frac{EA}{l}.
\]
A more realistic situation is one where the V-shaped beam experiences both a temperature increase \( \Delta T \) applied to actuate the device as well as an external force \( F \) in reaction to the displacement. This displacement is,
\[
U_y^{\Delta T + F} = U_y^{\Delta T} + U_y^F = \frac{2\alpha \Delta T E A s + F}{K_{tb}}.
\]
In the MEMS-based tensile loading device, a number of heat sink beams running between the shuttle and substrate are placed near the specimen to reduce the influence of the thermal actuator on the temperature of the specimen (see for example Fig. 1.8). Each pair of heat sink beams has a stiffness in the direction of the shuttle motion of

\[
K_{sb} = 2 \frac{12EI_{sb}}{l_{sb}^3} = \frac{2Eb_{sb}^3h}{l_{sb}^3},
\]

where \( I_{sb} \), \( l_{sb} \), and \( b_{sb} \) are the moment of inertia, length, and width of the heat sink beams respectively.

Finally, combining all these factors to make a thermal actuator with \( m \) pairs of thermal beams and \( n \) pairs of heat sink beams, the total stiffness and shuttle displacement are:

\[
K_{TA} = mK_{tb} + nK_{sb},
\]

\[
U_{TA} = \frac{U^{\Delta T}mK_{tb} + F}{K_{TA}} = \frac{2m\alpha\Delta TEAs + F}{K_{TA}},
\]

where \( U^{\Delta T} \), given by (1.5), is the displacement of the actuator in the absence of heat sink beams. Here the relation for the displacement is obtained by imposing compatibility in the kinematics of the systems of the thermal and heat sink beams.

1.3.3.3. Thermomechanical response of entire loading device

![Fig. 1.7. Lumped model of the entire tensile loading device with internal forces and displacements shown in free body form.](image)

With the mechanical response of the thermal actuator known for a given current input, it is now possible to formulate a set of equations governing the behavior of the entire device. A lumped model of the entire device
is constructed as shown in Fig. 1.7. Here $K_S$ is the stiffness of the tensile specimen, $K_{LS}$ is the stiffness of the load sensor corresponding to the folding beams by which it is suspended, $K_{TA}$ is the stiffness of the thermal actuator computed in (1.6), and $U_{LS}$ is the displacement of the load sensor. The central shuttle is assumed to be rigid.

The governing equations for the lumped system shown in Fig. 1.7 are given by:

$$
\Delta U_S = U_{TA} - U_{LS}
$$

$$
U_{TA} = \frac{2m\alpha \Delta T E A_s - F_{TA}}{K_{TA}}
$$

$$
F_{TA} = F_S = F_{LS}
$$

$$
F_S = K_S \Delta U_S
$$

$$
F_{LS} = K_{LS} U_{LS},
$$

where $s = \sin \theta$ and $\Delta U_S$ is the elongation of the specimen. Solving the system (1.8), the displacement of the thermal actuator $U_{TA}$, the tensile force on the specimen $F_S$, the elongation of the specimen $\Delta U_S$, and the corresponding displacement of the load sensor $U_{LS}$ are obtained:

$$
U_{TA} = \frac{2m\alpha \Delta T E A_s}{K_{TA} + K_{TA}K_{LS}/K_S + K_{LS}} + \frac{2m\alpha \Delta T E A_s}{K_{TA} + K_S + K_{TA}K_S/K_{LS}}
$$

$$
F_S = \frac{2m\alpha \Delta T E A_s}{K_{TA}/K_S + 1 + K_{TA}/K_{LS}}
$$

$$
\Delta U_S = \frac{2m\alpha \Delta T E A_s}{K_{TA} + K_S + K_{TA}K_S/K_{LS}}
$$

$$
U_{LS} = \frac{2m\alpha \Delta T E A_s}{K_{TA} + K_{TA}K_{LS}/K_S + K_{LS}}.
$$

These represent the critical parameters in obtaining force-displacement data using the MEMS-based tensile loading device.

1.3.4. Multiphysics FEA of the thermal actuator

While the displacement of the actuator in vacuum is easily characterized experimentally, the temperature distribution is more difficult to obtain. Therefore a coupled-field simulation is particularly necessary. This analysis also helps to assess the temperature at the actuator-specimen interface and to examine the effectiveness of the thin heat sink beams in controlling the temperature increase of the specimen during actuation.
The MEMS-based tensile stage is intended to operate within the SEM or TEM. Thus the following finite element electrothermal analysis is carried out for the case where the device operates in vacuum. The actuation voltage applied across the anchor points serves as the input while the output includes both the actuator temperature and displacement fields. Displacements at the anchor points are held fixed in the mechanical boundary conditions. The thermal boundary conditions are zero temperature change at the anchors.

![Diagram](image)

Fig. 1.8. (a) Temperature increase (°C) and (b) displacement field (nm) in the thermal actuator. The displacement component plotted is in the shuttle axial direction. (c) Temperature (°C) and (d) displacement field (nm) in the thermal actuator with three pairs of heat sink beams at the specimen end. In this analysis, the heat sink beams are 40 µm in length and 4 µm wide with 16 µm spacing between them. ANSYS Multiphysics, version 6.1 was used in this analysis.

Fig. 1.8a,b depicts the temperature and displacement in the thermal actuator for an actuation voltage of 1 V. As previously described, heat dissipation through the anchors is the dominant dissipation mechanism. Since the shuttle is furthest from the anchors, the highest temperature occurs in the shuttle. Due to the nonuniformity of the temperature distribution, the displacement is also nonuniform.
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Heating of the specimen during actuation is unavoidable as a result of the increased temperature of the shuttle to which the sample is attached. However, this effect is minimized with the addition of a series of heat sink beams running between the shuttle and substrate near the shuttle-specimen interface as shown in Fig. 1.8c,d. To avoid out-of-plane bending, another three pairs of heat sink beams are placed at the opposite end of the shuttle. Comparing this to the case without the heat sink beams (Fig. 1.8a,b), this configuration allows for more than twice the displacement at the specimen end of the shuttle for the same allowable temperature increase at the shuttle-specimen interface. The problem of specimen heating can be further mitigated with the addition of a thermal isolation layer between the actuator and specimen following the custom microfabrication process\(^38\) for highly temperature-sensitive samples.

1.3.5. **Buckling analysis**

For large tensile loads, buckling of the inclined beams in the thermal actuator is a concern. This occurs in the plane of minimum moment of inertia when the internal force exceeds the critical buckling load. Depending on the beam dimensions, this plane can be either parallel or orthogonal to the surface of the substrate. In this analysis, each beam is assumed to be fixed at the end where it is anchored to the substrate, while it is able to translate along the shuttle’s axial direction with no rotation at the other end. The critical axial force at which buckling occurs is:

\[
P_{cr} = \frac{\pi^2 EI_{\text{min}}}{l^2}.
\]

For an unloaded thermal actuator (disconnected from the sample and any heat sink beams), the axial internal force is \(R_{\Delta T} = \alpha \Delta TEA \frac{\phi^2}{x^2 + \gamma^2}\) where \(R_{\Delta T}\) is given by (1.4). For an actuator connected to a tensile sample, heat sink beams, and load sensor, the maximum possible axial force achieved is \(\alpha \Delta TEA\), i.e. when the actuator is attached to an elastic system of infinite stiffness and cannot translate. The true axial force experienced by the beams during operation falls somewhere between the two extremes.

1.3.6. **Evaluating the analytic and finite element models**

The thermal actuator is calibrated experimentally to verify the analytical and FEA models described above. Fig. 1.9\(^25\) shows a comparison the analytical and FEA predictions of actuator displacement for a given current
input with experimentally-measured results. The displacement of the actuator was measured in the SEM,\textsuperscript{26} giving spatial resolution of better than 5 nm. Using the analytical model, the displacement is computed based on experimentally-measured temperatures in the actuator.\textsuperscript{25} As mentioned in Sec. 1.3.4, the input to the multiphysics model is the voltage applied across the anchor points of the thermal actuator. In order to obtain the resulting current, the resistance of the actuator is computed using the output temperature and a value of resistivity corresponding to the average temperature of the device.\textsuperscript{25}

The models agree well with the experimentally-measured actuator displacements as shown in Fig. 1.9. This suggests the models are useful in predicting the behavior of thermal actuators of other geometry. At large currents (above approximately 12 mA), both the analytical and FEA models deviate slightly from the experimental results. This can be explained largely by inaccuracies in material parameters such as resistivity and thermal conductivity at high temperature.\textsuperscript{29,35} Furthermore, the microstructure of polysilicon begins to be modified at these high current levels and elevated temperatures.\textsuperscript{25}
1.3.7. **Load sensor**

The load sensor consists of a differential capacitive displacement sensor suspended on a set of elastic members of known stiffness. By calibrating the stiffness of the sensor\(^{26,40}\) the load is computed based on the measured displacement. The differential capacitive displacement sensor\(^{41-43}\) is chosen for its sensitivity and linear behavior over a range of displacements appropriate for tensile testing of nanostructures.

![Simple model of the differential capacitor](image1.png)

**Fig. 1.10.** (a) A simple model of the differential capacitor. (b) Double chip architecture used for measuring capacitance change. The capacitance change is proportional to the output voltage change.

The differential capacitive sensor is comprised of a movable rigid shuttle with electrodes (or “fingers”) pointing outward as shown schematically in Fig. 1.10.\(^{26}\) These fingers are interdigitated between pairs of stationary fingers (Fig. 1.10b) fixed to the substrate. Under no load, each movable finger sits centered between the two stationary fingers. Each set of fingers (one movable and a stationary on either side) forms two capacitors, one between the movable finger and each stationary finger. The entire capacitance sensor is equivalent to two combined capacitances, \(C_1\) and \(C_2\), as
shown in Fig. 1.10a, namely,

\[ C_1 = C_2 = C_0 = \epsilon N A d_0 (1 + f), \]

where \( \epsilon \) is the electric permittivity, \( N \) is the number of movable fingers, \( A \) and \( d_0 \) are the area of overlap and initial gap respectively between the movable finger and each stationary finger, and \( f = 0.65d_0/h \) is the fringing field correction factor with \( h \) being the beam height.\(^{44} \)

The movable fingers are attached to the folded beams via the rigid movable shuttle so their displacements are equivalent. This displacement yields a change in capacitance given by,

\[ \Delta C = C_1 - C_2 = N \epsilon A \left( \frac{1}{d_0 - \Delta d} - \frac{1}{d_0 + \Delta d} \right) \approx \frac{2 N \epsilon A}{d_0^2} \Delta d, \]

where \( \Delta d \) is the displacement of the load sensor. Note the fringing effect factor cancels. For displacements \( \Delta d \) within 50\% of the initial gap \( d_0 \), the capacitance changes approximately linearly with the sensor displacement. This relatively large range of linear sensing is a major advantage of differential capacitance sensing over direct capacitance sensing which uses a single fixed beam for each movable beam.

A variety of circuit configurations may be used in measuring capacitance.\(^{41,42} \) Fig. 1.10 shows schematically the charge sensing method used in the device described in this chapter. This method mitigates the effects of parasitic capacitances that generally occur in electrostatic MEMS devices. Here the change in output voltage \( \Delta V_{\text{sense}} \) is proportional to the capacitance change,

\[ \Delta V_{\text{sense}} = \frac{V_0}{C_f} \Delta C, \]

where \( V_0 \) is the amplitude of an AC voltage signal applied to the stationary fingers and \( C_f \) is the feedback capacitor shown in Fig. 1.10.

Minimizing stray capacitance and electromagnetic interference is critical in high resolution capacitance measurements. In this case, integrating the MEMS differential capacitor and sensing electronics on a single chip would minimize these effects, allowing detection of changes in capacitance at the atto-Farad level.\(^{42} \) However, this would greatly increase fabrication complexity. The double chip architecture depicted in Fig. 1.10 is an alternative to the single chip scheme. Here the MEMS-based system is fabricated on one chip while a commercial integrated circuit chip (for example, Universal Capacitive Readout MS3110, Microsensors, Costa Mesa, CA) is used to
measure changes in capacitance. Both chips are housed on a single printed
circuit board.

An equivalent circuit for the entire device, including the electrothermal
actuator, is shown in Fig. 1.11. The load sensor shuttle is connected elec-
trically to the substrate through the anchor points. The capacitances are given by,

\[ C_1 = N \epsilon \left( \frac{A_1}{d_0 + \Delta d} + \frac{A_2}{g} + 0.65 \frac{A_1}{h} \right), \quad (1.11) \]
\[ C_2 = N \epsilon \left( \frac{A_1}{d_0 - \Delta d} + \frac{A_2}{g} + 0.65 \frac{A_1}{h} \right), \quad (1.12) \]
\[ C_3 = N \epsilon \left( \frac{A_1}{d_3} + 0.65 \frac{A_1}{h} \right), \quad (1.13) \]

where \( N \) is the total unit number of differential capacitors, \( A_1 \) is the overlapping area of the stationary finger with the movable finger, \( A_2 \) is the overlapping area of the stationary finger with the substrate, \( d_0 \) is the gap between the stationary finger and the movable finger, \( \Delta d \) is the displacement of the movable finger, \( d_3 \) is the gap between the two stationary fingers, \( g \) is the gap between the fixed finger and substrate, and \( h \) is the finger thickness. In (1.11) and (1.12), the first term represents the capacitance between each fixed finger and the corresponding movable finger. The second term is the capacitance between the fixed finger and the shield beneath the load sensor which is held at the same potential as the movable finger. The third term considers the fringe effect. Note that when \( \Delta d = 0 \), \( C_1 = C_2 = C_0 \).

The capacitance term from the fringe effect \( C_3 \) cannot be neglected as \( d_3 \) is typically comparable to \( d_0 \). Fig. 1.11b shows the circuit used for the device in Fig. 1.11a while Fig. 1.11c shows the equivalent circuit due to a \( \Delta - Y \) transformation. Here the equivalent capacitances are given by,

\[ C_{13} = \frac{C_1 C_2 + C_2 C_3 + C_3 C_1}{C_2} \]
\[ C_{23} = \frac{C_1 C_2 + C_2 C_3 + C_3 C_1}{C_1} \]
\[ C_{12} = \frac{C_1 C_2 + C_2 C_3 + C_3 C_1}{C_3} \]  \( (1.14) \)

Combining (1.11-1.13) and (1.14) gives the difference in capacitance,

\[ C_{13} - C_{23} = N \epsilon_0 A_1 \left( 1 + \frac{C_1 + C_2}{C_1 C_2} \right) \left( \frac{1}{d_0 + \Delta d} - \frac{1}{d_0 - \Delta d} \right) \]
\[ \approx 2 N \epsilon_0 A_1 \frac{1 + 2C_3/C_0}{d_0^3} \Delta d, \quad (1.15) \]

\[ C_{12} = C_1 + C_2 + \frac{C_1 C_2}{C_3} \approx 2C_0 + \frac{C_0^2}{C_3} = constant. \quad (1.16) \]
These capacitance values agree well with those obtained in FEA simulation. The quantity $C_{13} - C_{23}$ in (1.15) is the capacitance that is measured experimentally while using the MEMS tensile loading device. This is approximately five times greater than the quantity $C_1 - C_2$, which is advantageous in measuring sub-femtoFarad capacitances. In practice, the displacement-capacitance relation is calibrated experimentally for improved accuracy.

1.4. Design criteria

Taking into consideration the above analyses, the following design criteria are set to achieve an effective and reliable material testing system:

(1) Large load sensor displacements to maximize load resolution;
(2) Low temperature at the actuator-specimen interface to avoid artificial heating of the specimen;
(3) The testing system operates as a displacement control, i.e., the stiffness of the thermal actuator is significantly higher than that of the specimen and load sensor, and;
(4) The actuator does not buckle within the operational temperature range.

The specimen stiffness, failure load, and elongation at failure ($\Delta U_S$) dictate the choice of actuator geometry and the number and dimensions of the beams. Consequently, optimization of the device design requires some preliminary knowledge of the specimen behavior as is customary in experimental mechanics.

In choosing the stiffness of the load sensor ($K_{LS}$), a compromise between the maximum force applied to the specimen and sensor displacement must be reached. The force applied to the specimen is $F_S = F_{LS} = K_{LS}U_{LS}$. For a differential capacitance load sensor, the displacement resolution remains approximately constant. Thus the smaller the stiffness, the greater the load resolution. However, in order to achieve the required elongation of the sample $\Delta U_S$ for failure, the displacement of the thermal actuator $U_{TA}$ must increase. This in turn requires higher temperatures resulting in greater heating of the specimen.

The displacement of the thermal actuator (unconstrained by heat sink beams or a specimen) depends on the beam length $l$, beam angle $\theta$, temperature increase, and stiffness ratio $\Psi$ given in (1.5). The longer the beams, the greater the displacement. However, longer beams are more likely to stick to the substrate during microfabrication and are more prone to buck-
Thus there is a practical limit placed on the length of the inclined beams.

Fig. 1.12. Important parameters in the device design as functions of the thermal beam angle: (a) displacement; (b) stiffness of the thermal actuator; and (c) internal stress. The parameters in (a) and (b) are plotted as dimensionless quantities. Beam dimensions of length $l = 300 \mu m$, width $b = 8 \mu m$ and height $h = 3.5 \mu m$ were chosen.

Fig. 1.12a shows the displacement of the thermal actuator as given by (1.5) as a function of the beam angle. Here the displacement is plotted as a dimensionless quantity $(U\Delta T/\alpha \Delta Tl)$ for a fixed stiffness ratio of $\Psi = 1406$. The displacement increases with a decrease in the beam angle in the range $\theta > 2^\circ$. Thus to obtain a given displacement, an actuator with a smaller beam angle ($\theta > 2^\circ$) requires a smaller temperature increase and equivalently a smaller actuation voltage.

Regarding the third design criterion, it is desirable to have the thermal actuator operate in displacement control mode. The ability to prescribe the displacement of the tensile specimen is critical in mechanical testing. This allows important mechanical phenomena such as stress softening and fracture to be captured. In order to achieve true displacement control, the actuator would ideally have infinite stiffness, allowing it to reach the desired displacement regardless of the required force (in theory, it could apply infinite force if given infinite stiffness). In practice, the actuator stiffness must be significantly larger than that of the specimen and load sensor. Fig. 1.12b plots the stiffness of the thermal actuator as a function of beam angle for the same fixed stiffness ratio ($\Psi = 1406$). Again the quantities are plotted in dimensionless form with the dimensionless stiffness being $K_{tb}l/2EA$. Contrary to the inverse relationship between actuator displacement and beam angle, the actuator stiffness increases with beam angle. Thus there is a trade off between maximizing the displacement and stiffness of the actuator.
The final design criterion considers the possibility of buckling. As the temperature of the beams rises, the internal forces build, increasing the possibility of buckling. Fig. 1.12c plots the internal axial force as a function of the beam angle as well as the critical minimum buckling force for a temperature increase of $\Delta T = 800$ K. Since the recrystallization temperature of polysilicon (the material of the beams) is approximately 800 K, the beams are not expected to buckle within the functional temperature range. The plot shows that the actuator buckles when the beam angle is less than approximately 5° at 800 K.

![Fig. 1.13. Two types of thermal actuators for testing various types of nanostructures. (a) 10 pairs of thermal beams with beam angle of 10°; (b) 5 pairs of thermal beams with a beam angle of 30°](image)

To summarize, an actuator with a small beam angle requires the lowest temperature increase for a given displacement. However, the structural stability of the actuator decreases with beam angle. For specimens requiring large actuator displacements, a beam angle of 10° may be selected (Fig. 1.13a). For specimens requiring only moderate displacements and greater forces, a beam angle of 30° is more appropriate (Fig. 1.13b). In each case, the number of thermal beams is chosen to achieve the desired actuator stiffness-to-load sensor stiffness and actuator stiffness-to-specimen stiffness ratios. Likewise, the load sensor stiffness is chosen according to (1.8) and (1.9) once an estimate of the specimen stiffness and elongation at failure is made.

### 1.5. Material testing

The MEMS-based tensile stage can be used to test nanometer-scale materials and structures ranging from nanowires and nanotubes to ultra-thin films. As the structures shrink below the sub-micron and into the nanometer scale, new mechanisms dominate their mechanical behavior. For in-
stance, nanowires and nanotubes possess a relatively large surface area-to-volume ratio. Consequently, interfaces, interfacial energy, and surface topography play an increasingly important role in their deformation and failure processes. In larger structures, generation and motion of dislocations dictates material behavior. As grain sizes or structural dimensions fall below 50 to 100 nm, surface and intermolecular mechanisms gain influence over material behavior. Therefore understanding the mechanics of these new materials and structures is essential. As an example, the following briefly demonstrates the use of the MEMS-based tensile stage for in-situ SEM and TEM of testing nano-scale polysilicon films, palladium nanowires, and multi-walled carbon nanotubes.

1.5.1. Sample preparation

![Image](image.png)

Fig. 1.14. Sample preparation. (a) A polysilicon thin film tensile specimen cofabricated with the MEMS device and further thinned by FIB machining. (b) A palladium nanowire being manipulated into place on the MEMS device using a tungsten probe and nanomanipulator.

The size and fragile nature of nano-scale materials and structures demands specialized techniques for preparation and mounting on the MEMS device. Thin films may be cofabricated with the MEMS device. This eliminates any handling. For example, freestanding polysilicon films were cofabricated with the MEMS device between the actuator and the load sensor (Fig. 1.14a).

Due to limitations in the resolution of the photolithography used to make the devices, the initial film thickness could not be made thinner than approximately 2 μm. To reduce the thickness dimension, the polysilicon specimen is further machined by focused ion beam (FIB) down to 350-450 nm.
Individual nanowires and nanotubes may either be grown across the gap between the actuator and load sensor or placed by nanomanipulator as shown in Fig. 1.14b. This procedure involves use of a nanomanipulator operated within an SEM to pick up and place an individual nanostructure across the gap, followed by electron beam-induced deposition (EBID) of platinum to weld the ends of the structure in place.

1.5.2. Tensile tests

1.5.2.1. Tensile tests of polysilicon thin films

![Stress-strain data for (a) a polysilicon thin film specimen and (b) a palladium nanowire.](image)

Fig. 1.15. Stress-strain data for (a) a polysilicon thin film specimen and (b) a palladium nanowire.

Thin film specimens cofabricated with the MEMS device were tested in the SEM. The results of a tensile test of a polysilicon specimen prepared as described in 1.5.1 are shown in Fig. 1.15a. Here the stress-strain curve shows strong linearity with a Young’s modulus of $156 \pm 17$ GPa. This result is consistent other reported values for polysilicon films.

1.5.2.2. Tensile tests of nanowires

The tensile test of a palladium nanowire, shown in Fig. 1.15b, reveals an interesting point. The nanowire was stressed to 1.5 GPa, significantly higher than the yield stress of bulk nanocrystalline Pd, and remained elastic without fracture. This phenomenon, which is attributed to the high stress threshold for the nucleation of defects, confirms that the strength of the material increases as its scale decreases. Ultimately, the
strength should tend to approach the theoretical strength of the material (approximately 1/10 of its Young’s modulus).  

1.5.2.3. Tensile tests of carbon nanotubes and the effects of irradiation

In-situ SEM and TEM tensile tests of multi-walled carbon nanotubes using the MEMS device allow insight into their failure mechanisms. Fig. 1.16 shows sequential SEM images of the in-situ SEM tensile testing of a MWNT. In this case, fracture occurs in a typical “sword-in-sheath” fashion. It is possible that the outermost shell breaks and subsequently the inner concentric shells telescope out. However, the instrument resolution does not permit identification of single or multiple shell failure. As described in 1.5.1, the two ends of the outermost shell are clamped to the testing device using EBID of platinum. Consequently, it is reasonable to assume that the outermost shell carries the load and breaks under tensile loading as only van der Waals interactions between the concentric inner shells are expected. Based on the assumption that there is minimal load transfer between the outermost shell and subsequent inner shells, stress and strain are often calculated using the measured outer diameter of the MWNT and an assumed shell thickness of 0.34 nm (equal to the interlayer distance of 0.34 nm), as opposed to using the combined thickness of multiple concentric shells.

In contrast to tests of unmodified MWNTs, in-situ TEM tensile tests of MWNTs exposed to high-energy electron or ion beam irradiation reveal that multiple shells or the entire cross section break simultaneously, resulting in greater stiffness. This observation suggests that the irradiation introduces crosslinks between shells, resulting in load transfer. A series of tests demonstrating this effect were performed using the MEMS device on MWNTs exposed to varying degrees of ion or electron beam radiation, as well as on unexposed MWNTs as a control. These tests are summarized in Fig. 1.17 and Tab. 1.1. Here it is important to note that the irradiation energy of the electron beam used for SEM imaging during mounting of the MWNTs on the MEMS device is well below the threshold for atomic structure modification and thus is assumed to have negligible effect relative to the high-energy electron and ion beam exposure.

MWNTs exposed to ion irradiation demonstrated significantly greater stiffness than those that were not. In Test 1 (see Tab. 1.1), the MWNT was irradiated with Ga+ ions at a flux of $10^{13}$ e cm$^{-2}$s$^{-1}$ for 10 seconds.
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Fig. 1.16. Sequential SEM images and corresponding stress-strain data of a tensile test of a multi-walled carbon nanotube. The nanotube is 2 μm in length and 42 nm in diameter.
Table 1.1. Irradiation conditions and measured mechanical properties of six MWNTs.\(^{40}\)

<table>
<thead>
<tr>
<th>Test #</th>
<th>Gauge Length [\mu m]</th>
<th>Outer Dia. [nm]</th>
<th>Ion Radiation Dose [e \text{ cm}^{-2}]</th>
<th>Electron Radiation Dose [e \text{ cm}^{-2}]</th>
<th>Breaking Force [\mu \text{N}]</th>
<th>$\Delta L$ [nm]</th>
<th>$\sigma_2$ [GPa]</th>
<th>$E^a$ [GPa]</th>
</tr>
</thead>
<tbody>
<tr>
<td>1</td>
<td>6.69</td>
<td>191</td>
<td>$10^{14}$</td>
<td>-</td>
<td>31.1</td>
<td>195.6</td>
<td>152.5</td>
<td>5.200</td>
</tr>
<tr>
<td>2</td>
<td>3.02</td>
<td>142</td>
<td>$0.5 \times 10^{14}$</td>
<td>-</td>
<td>20.4</td>
<td>101.2</td>
<td>134.5</td>
<td>4.000</td>
</tr>
<tr>
<td>3</td>
<td>2.85</td>
<td>169</td>
<td>-</td>
<td>$4.5 \times 10^{14}$</td>
<td>9.9</td>
<td>143.8</td>
<td>54.8</td>
<td>1.100</td>
</tr>
<tr>
<td>4</td>
<td>3.30</td>
<td>108</td>
<td>-</td>
<td>$1.5 \times 10^{14}$</td>
<td>4.3</td>
<td>97.5</td>
<td>37.3</td>
<td>1.300</td>
</tr>
<tr>
<td>5</td>
<td>3.82</td>
<td>96</td>
<td>-</td>
<td>-</td>
<td>1.9</td>
<td>221.9</td>
<td>18.5</td>
<td>300</td>
</tr>
<tr>
<td>6</td>
<td>2.06</td>
<td>42</td>
<td>-</td>
<td>-</td>
<td>2.1</td>
<td>156.8</td>
<td>48.5</td>
<td>1,000</td>
</tr>
</tbody>
</table>

\(^a\)Stress and strain are computed under the assumption that only the outermost shell of the MWNT bears the load (i.e., the cross-sectional area is taken to be that of the outermost shell alone).

using a 30 kV accelerating voltage. An in-situ TEM tensile test was then performed using the MEMS device. The corresponding inset of Fig. 1.17 shows clearly that the entire cross section broke as opposed to the typical telescoping, “sword-in-sheath” mechanism. Thus stress and modulus values reported in Tab. 1.1, which were computed assuming only the outermost shell to be load bearing, appear meaningless in this case. In fact, ion irradiated specimens exhibit values of Young’s modulus of 5,200 and 4,000 GPa which are significantly higher than those reported elsewhere in the literature based on quantum mechanics calculations.\(^{53–55}\) Note also that Ding et al.\(^{13}\) reported moduli ranging from 620 to 1,200 GPa based on the assumption that only the outermost shell is load bearing. However, they do not provide evidence that only the outer shell failed.

MWNTs exposed to electron beam irradiation also showed greater stiffness than the unexposed sample, although to a lesser degree. In Test 4 (see Tab. 1.1), the MWNT was exposed to electron beam radiation with a flux of $1.5 \times 10^{19}$ e cm\(^{-2}\)s\(^{-1}\) for 100 seconds within the TEM at an acceleration voltage of 200 kV. In this case, the failure was telescopic in nature as shown in the corresponding inset of Fig. 1.17. However, high-resolution TEM images show that more than one shell broke simultaneously rather than only the outermost shell. Thus again, the computed values of stress and strain based on the outermost shell assumption, do not accurately represent the true material behavior.

In summary, tensile tests of MWNTs exposed to high-energy electron or
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Fig. 1.17. Force-displacement data measured for multi-walled carbon nanotubes exposed to varying degrees and types of radiation. Corresponding irradiation conditions and test parameters are summarized in Tab. 1.1.

ion beam irradiation reveal changes in the mode of failure, suggesting that the irradiation introduces crosslinks between shells. Others have reported corroborating evidence based on experiments (for single-walled carbon nanotube bundles) and first principle calculations.\textsuperscript{56–58} Above a certain energy threshold, electron and ion beams can produce vacancies in the nanotube shells and corresponding interstitials in the inter-shell spacing.\textsuperscript{56} Moreover, molecular dynamics simulations revealed that these interstitial atoms can form stable and covalent bonds between shells.\textsuperscript{57} These simulations further demonstrated that the development of covalent bonds under moderate beam irradiation can increase the failure strength of MWNTs while excessive irradiation degrades the mechanical properties due to structural damage (cluster of vacancies) and/or amorphization.\textsuperscript{57,58} The experimental observations reported here agree strongly with the predictions of these simulations.

These findings show that both electron and ion irradiation could be used to enhance the mechanical properties of MWNTs. However, in light of the observed changes in failure mechanisms, conclusions based on existing stress and strain data computed under the assumption that only the
outermost shell bears the load should be drawn with caution. The observations reported here suggest there is some degree of load sharing between shells. For this reason, conclusions inferred from the data reported here focus on the stiffness of the nanostructures, which were measured directly. With advances in TEM image acquisition, the MEMS-based testing technique reported here should allow direct imaging of the evolution of the failure and the number of shells failing simultaneously.

1.6. Summary

Mechanical characterization of nanometer-scale materials and structures presents a unique set of challenges. The excellent force and displacement resolution of MEMS make them ideal components for material characterization. This chapter presented the modeling and analysis involved in the design of a MEMS-based material testing system allowing simultaneous load-displacement measurement combined with real-time SEM or TEM imaging of the specimen. This system uses a thermal actuator to apply a tensile load and a differential capacitance displacement sensor of known stiffness to determine the applied load. An analytical model of the thermal actuator involved an electrothermal analysis to determine the temperature distribution in the actuator, followed by a thermomechanical analysis to determine the resulting displacement. A coupled-field finite element analysis confirms the analytical model. The differential capacitive load sensor was analyzed to determine the output voltage for a given displacement. A set of design criteria were established based on the analyses as guidelines for design of similar devices. Finally, examples of application of the MEMS-based material testing system to polysilicon thin films, palladium nanowires, and multi-walled carbon nanotubes were presented. Tensile tests of palladium nanowires demonstrated a major strength increase as compared to bulk palladium. Tensile tests of multi-walled carbon nanotubes exposed to varying degrees of electron and ion beam irradiation showed differences in failure mechanisms and an increase in stiffness with the level of irradiation. This is attributed to the formation of crosslinks between shells.

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